Recent Advances in Doped Bi2O³ and its Photocatalytic Activity: A Review

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ABSTRACT

Bismuth Oxide $(Bi₂O₃)$ has a very promising photocatalytic ability to degrade waste pollutants under visible light irradiation because it has a small energy gap of around 2.85-2.58 eV. Although it has excellent potential as a photocatalyst, $Bi₂O₃$ has the disadvantage of a high electron-hole pair recombination rate, which will reduce its photocatalytic activity. To overcome these problems, surface modifications, defect recognition, or doping of $Bi₂O₃$ are carried out to obtain a more effective and efficient photocatalyst to degrade waste pollutants under visible light irradiation. Several studies by researchers have been described for the modification of $Bi₂O₃$ by doping. Various types of doping are given, such as doping in elements or doping in the form of compounds to form composites. Based on several studies that have been described, appropriate doping has been shown to increase the photocatalytic activity of $Bi₂O₃$.

Keywords: Bi2O3, Photocatalyst, Doping

INTRODUCTION

The growth of the modern industry is getting faster in recent years has led to an increase in environmental pollution, especially water pollution. Pollution of the water environment is a severe problem and must be solved because the water environment is one of the sources of life for living things. If the water environment is polluted, it will threaten the ecosystem and the health of living things¹. Water pollution from industrial and medical waste is a source of considerable pollution in water environmental problems if the waste is discharged directly into the river without any special treatment. The waste's toxic, accumulating, and potentially carcinogenic nature poses a severe threat to the aquatic environment. Therefore, an effective and efficient method is needed to overcome this water pollution problem²⁻⁴

Advanced Oxidation Process (AOPs) has been widely used as an environmentally friendly, green process method, and cost-effective so that it is more productive in providing solutions to water environmental problems. AOPs involve the generation of hydroxyl radicals in sufficient quantities to affect water purification and utilize the oxidation reaction process⁵. Photocatalytic technology shows good prospects for use in this technique because it can carry out oxidation reactions and generate hydroxyl radicals and be sustainable and environmentally friendly⁶.

Semiconductor photocatalysts have been widely applied to overcome the problem of water pollution. Several semiconductors such as $TiO₂$, ZnS, and ZnO have been widely applied to degrade pollutants in the aquatic environment under Ultra Violet (UV) light irradiation. However, due to the amount of UV light from solar radiation is less than visible light, it is crucial to take advantage a widely used of visible light. So it is necessary to do further research for other photocatalysts that are more efficient and able to work in visible light⁷. As an attractive photocatalytic material, Bismuth oxide $(Bi₂O₃)$ has been widely studied for its easy synthesis, controllable energy gap, and high visible light response⁸⁻⁹.

 $Bi₂O₃$ is a p-type metal oxide semiconductor with a narrow energy gap of 2.85-2.58 eV, which can be widely used as a photocatalyst in visible light⁹. . The characteristics of $Bi₂O₃$ are auspicious for application as a photocatalyst such as high redox reversibility, significant photoluminescence, refractive index, photoconductivity, and dielectric permittivity, as well as low resistivity 10 . There are five crystallographic polymorphs of Bi2O3, namely (monoclinic), (tetragonal), (cubic bcc), (cubic fcc), and (orthorhombic) 11 . The phase change cycle of $Bi₂O₃$ can be seen in [Fig 1].

Fig 1: Bi2O³ phase transformation

Increasing the temperature in the α -Bi2O³ phase°C to 730°C obtained the δ- $Bi₂O₃$ phase, which is stable to the melting point at 824° C, cooling the δ -Bi₂O₃ phase at 646°C produces the β-Bi₂O₃ phase or γ-Bi₂O₃ phase at 639°C. The β-Bi₂O₃ phase changed to the α -Bi₂O₃ phase on cooling to 303 \degree C, and the γ-Bi₂O₃ phase at 500 \degree C. The transition to the γ -Bi₂O₃ phase can also occur by cooling δ -Bi₂O₃ at 635-640^oC. The α-monoclinic and γ-cubic bcc phases are semiconductors, while the β-tetragonal and γ-cubic fcc are excellent conductors of oxide ions and can be considered aniondeficient fluorite structures bismuth occupies the fcc site, undergoing oxygen sublattice defects. The γ -Bi₂O₃phase can stand up to room temperature with a slow cooling rate. The highest conductivity occurs in the δ -Bi₂O₃ phase¹²⁻¹³. There is a stable phase at low temperature, namely α - $Bi₂O₃$, and a stable phase at high temperature, namely $\delta - \text{Bi}_2\text{O}_3$; the other three are metastable phases 14 .

Some of the characteristics of $Bi₂O₃$ described have the potential to make it a photocatalyst. However, using pure $Bi₂O₃$ as a photocatalyst has some drawbacks, such as relatively low photocatalytic activity and a high electron-hole pair recombination rate. To overcome this problem, it is necessary to modify the surface, identify defects, or doping $Bi₂O₃$ to obtain a more effective and efficient photocatalyst to degrade waste pollutants under visible light irradiation. Several researchers have modified $Bi₂O₃$ by doping. The types of doping given vary, such as doping elements and compounds to form composites. Several studies will be discussed in this journal for the doped and synthesized $Bi₂O₃$ photocatalyst using the precipitation method. Precipitation is one method of material synthesis that is relatively easy to do because it does not require complex technology, only mixing reagents with precipitating agents. In addition, it can be carried out at room temperature and produces higher yields 15 .

PHOTOCATALYST Bi2O³

Sonal, S. and R. Sharma¹⁶, have succeeded in synthesizing α -Bi₂O₃ doping Ni by the precipitation method. The X-Ray Diffraction (XRD) pattern obtained has a α - $Bi₂O₃$ phase based on JCPDS: 00-041-1449, which indicates no crystal structure deformation or phase change due to Ni doping. However, the existence of Ni doping causes the peak intensity decreased slightly, indicating a reduction in the crystallinity of the doped sample. The Nidoped $α$ -Bi₂O₃ sample experienced a slight shift in the value of 2θ towards the right to become more comprehensive. The Ni-doped α-Bi2O³ Scanning Electron Microscope (SEM) image shows a rod-like morphology with non-uniform or broken dimensions with an average 15–22 μm. The addition of Ni causes morphological changes in $α$ -Bi₂O₃ provide a rougher surface area due the structure forms broken and irregular rods to help the photoreaction process be better. The α -Bi₂O₃/Ni energy gap is lower than that of pure α-Bi₂O₃, which is 2.58 eV. The photocatalytic activity of Ni-doped $α$ -Bi₂O₃ samples was carried out by analyzing the photodegradation of Methylene Blue (MB), and showed an increase compared to pure α - $Bi₂O₃$. The color change ratio was up to 50% throughout 275 minutes before doping. After doping, there was an increase in the ratio, almost 81% MB degraded over the same time duration.

In addition, Meng, Q and Z. Ying¹⁷, have also succeeded in synthesizing Bi2O3/Ni with a Ni concentration of 1-5% (%mol) using the precipitation method. However, in this study, the addition of Ni caused a change in the crystalline phase of $Bi₂O₃$. XRD pure $Bi₂O₃$ diffraction pattern indicated as monoclinic α -Bi₂O₃ phase based on JCPDS: 71-0465. The addition of Ni 1% to $Bi₂O₃$ causes the peaks of α-Bi₂O₃ to disappear gradually, but several new prominent peaks appear and are confirmed as tetragonal $β$ -Bi₂O₃ phase according to JCPDS: 49-1762. When the Bi/Ni molar ratio is 2% , sample of α -Bi₂O₃ completely changes to tetragonal $β$ -Bi₂O₃. The crystal sizes of the 0-5% $Bi₂O₃/Ni$ are 53.5; 39.1; 35.7; 32.4; 32.8; and 41.8 nm. Doping Ni 3% has the smallest crystal size, its morphology shows a perfect microsphere structure. Increasing the concentration of Ni further displays a flower-like structure. The energy gap of 3% Bi₂O₃/Ni is 2.37 eV, and pure $Bi₂O₃$ is 3.06 eV. The photocatalytic activity was performed by degrading pyridine under visible light radiation for 60 minutes. The result was that the Bi_2O_3/Ni photocatalyst was much higher in performance than $Bi₂O₃$. The best

performance was observed in 3% Bi2O3/Ni with a degradation efficiency of around 93%.

Malathy, P and colleagues¹⁸, have succeeded in synthesizing $Bi₂O₃$ doped transition metals (Ni and Zn) by precipitation method. Based on the results, the XRD pattern of the sample showed that the crystal structure of $Bi₂O₃$ was not affected, but after doping with Ni and Zn the peak intensity changed. All diffraction peaks were formed entirely and indicated as tetragonal $β$ -Bi₂O₃ phase according to JCPDS: 65-1209. Due to transition metals doping, diffraction peaks did not appear due to higher dispersion and their low content. The average crystal sizes of β -Bi₂O₃ was 17 nm, $Bi₂O₃/Ni$ was 23 nm, and $Bi₂O₃/Zn$ was 20 nm. The results of the sample morphology analysis by SEM showed that β-Bi2O3, Bi2O3/Ni had a stem-like structure and $Bi₂O₃/Zn$ had a flower-like structure. The addition of transition metals causes this morphological change. The presence of Ni ions in $Bi₂O₃$ shows a significant redshift of the absorption peak towards the visible light region compared to $Bi₂O₃$ with energy gaps for β-Bi₂O₃, Bi₂O₃/Ni, and Bi₂O₃/Zn are 2.8; 2.69; and 2.74 eV respectively. The narrower energy gap of $Bi₂O₃/Ni$ can have photocatalytic activity in visible light, which is very good than $Bi₂O₃/Zn$ and $Bi₂O₃$. The degradation of Malachite Green (MG) under visible light irradiation with an irradiation duration of 180 minutes was used to evaluate the photocatalytic activity of $Bi₂O₃/Ni$. The results show that $Bi₂O₃/Ni$ has better photocatalytic activity than other photocatalysts. This corresponds to the energy gap obtained, which is the smallest.

The synthesis of $Bi₂O₃/Zn$ was also carried out by Viruthagiri, G. and colleagues^{19} , using a simple chemical precipitation method. Diffraction peaks of $Bi₂O₃$ indicated as monoclinic α -Bi₂O₃ phase according to JCPDS: 71-0465, the phase structure did not change after Zn doping. Pure $Bi₂O₃$ has a size of 57.2 nm, while after doping 1-5% Zn (%mol) the crystal size decreases to 42.94-54.56 nm. The pure $Bi₂O₃$ morphology shows many needle- or rod-shaped structures with sharp edges, whereas Bi2O3/Zn shows a wellisolated rod morphology with a porous structure and rough particle surface. The energy gap of pure $Bi₂O₃$ is 2.65 eV, while the energy gap of 1-5% $Bi₂O₃/Zn$ is around 2.76-2.68 eV. The photocatalytic activity of Bi2O3/Zn was investigated based on the degradation of Methylene Blue dye solution under visible light irradiation. The results showed that after 135 minutes of irradiation, Bi2O³ showed partial degradation of the dye, while the doped samples showed complete degradation. $Bi₂O₃/Zn$ showed a 95% higher degradation efficiency than pure $Bi₂O₃$.

In addition to Ni and Zn doping, Viruthagiri G and P. Kannan²⁰, conducted a study to synthesize cobalt (Co) doped $Bi₂O₃$ using the precipitation method. Doping was carried out with different Co concentrations (0.05-0.25 M). The XRD pattern of pure Bi2O³ nanoparticles was confirmed to have a monoclinic α-Bi2O³ crystal structure (JCPDS No. 71-0465). There are no peaks associated with cobalt or cobalt oxide in the $Bi₂O₃$ phase. However, the doping effect causes a non-monotonic shift in the diffraction pattern towards a wider 2θ region. The crystal size and lattice parameter values decrease gradually with Co doping. The surface morphology of pure $Bi₂O₃$ and $Bi₂O₃/Co$ (0.15 M) analyzed by Field Emission-Scanning Electron Microscope (FE-SEM) showed that $Bi₂O₃$ has a morphology like nanosheets, while $Bi₂O₃/Co$ (0.15 M) consists of many narrow nanoplates that are interconnected. Based on the UV-Vis Bi2O3/Co spectrum, as the concentration of Co doping increases, the absorption edge shifts towards a wider wavelength region, indicating a decrease in the energy gap of $Bi₂O₃$. The energy gap of $Bi₂O₃/Co$ (0.05-0.25 M) decreases with increasing dopant concentration from 2.21 eV to 1.94 eV, while pure $Bi₂O₃$ is 2.64 eV. The photodegradation of Methylene Blue dye solution under visible light irradiation for 135 min was carried out to evaluate the

photocatalytic activity of $Bi₂O₃$ and $Bi₂O₃/Co$ (0.15 M), and showed the best catalytic activity with a degradation efficiency of 97%, while pure $Bi₂O₃$ was only 76.15%.

Synthesis to produce a silver-doped Bi2O³ photocatalyst with a Bi/Ag molar ratio of 1-9% (%mol) by co-precipitation method was carried out by Li, Y and colleagues²¹. The XRD pattern shows that the composition of all samples is monoclinic α -Bi₂O₃ phase (JCPDS: 41-1449). In addition, the typical face-centered cubic (fcc) structure of Ag metal (JCPDS: 04- 0783) was indicated in the formation of pure silver with low crystallinity observed in the 9% Bi2O3/Ag XRD pattern. The Transmission Electron Microscope (TEM) analysis showed that the sample consisted mainly of high crystallinity nanosheets. $Bi₂O₃/Ag$ nanosheets show high absorption in the visible light region. The energy gap of $Bi₂O₃/Ag$ (2.59-2.25 eV) is lower than that of pure $Bi₂O₃$ (2.63 eV). The photocatalytic activity tested by degrading Methyl Orange (MO) under visible light irradiation. With duration off irradiation for 180 minutes, Ag 3% doped on $Bi₂O₃$ had the most optimal photocatalytic activity. The experimental results show that the rate of photocatalytic activity of Bi_2O_3/Ag is 3.45 times compared to pure $Bi₂O₃$. The photocatalytic activity $Bi₂O₃/Ag$ had better than pure $Bi₂O₃$, attributed to the Ag-doped nanosheet structure offering high electron-hole pair separation. Doping Ag was able to increase $Bi₂O₃$ activity, but excessive Ag doping causes a problem in the heterojunction structure acts as a charge carrier recombination center for electron and hole pairs. Defects and oxygen vacancies of the $Bi₂O₃$ lattice due to excessive Ag doping affect the decrease in photocatalytic activity of $Bi₂O₃/Ag$ samples (5-9%).

Liang, J and colleagues²², have synthesized Bi₂O₃/Fe porous microspheres. The results of the XRD pattern analysis showed that the crystal formed was a tetragonal β-Bi₂O₃ phase (JCPDS: 65-1209). $Fe³⁺$ ions undergo substitution to $Bi³⁺$ ions

in the Bi2O³ lattice structure but are not affected in the $Bi₂O₃$ crystal lattice. The SEM image shows that pure β -Bi₂O₃ has a porous microsphere structure with a diameter of about 7 μm. The microsphere structure consists of many Fe nanoparticles. UV-Vis spectra showed that Bi_2O_3/Fe had strong visible light absorbance in the wavelength range from 420 nm to 600 nm. The energy gap ranges from 2.25-1.67 eV. Increased absorption of visible light and decreased energy gap can increase the photocatalytic activity of β -Bi₂O₃. Photodegradation of Methyl Orange was tasted to determine the photocatalytic activity of $Bi₂O₃/Fe$. It was observed that the Methyl Orange dye molecule was able to degrade better using a $Bi₂O₃/Fe$ photocatalyst than pure β-Bi₂O₃under visible light irradiation with an irradiation duration of 60 minutes, the degradation rate of 0.0328 min-1 was obtained.

Sudrajat, H. et. al.²³, has conducted a study by applying two strategies simultaneously to improve Bi₂O₃ photocatalytic properties. The first strategy through carbon (C) and nitrogen (N) doping is simultaneously able to increase the absorption of visible light region, and the second strategy is Fe(III) grafting to increase the charge carrier separation. The synthesis results showed that $Bi₂O₃$ have diffraction peaks according to cubic $δ$ -Bi₂O₃ phase based on JCPDS: 52-1007. There did not indication of crystal phase and other impurities after C-N doping without Fe(III) grafting, but the intensity of diffraction peaks decreased and widened slightly, this may occur due to urea which inhibition of crystal growth during calcination. The crystal phase did not change after Fe (III) grafting, and the peak intensity were relatively similar. The existence of Fe (III) was only on the surface of $Bi₂O₃$ so that it did not change the microstructural of $Bi₂O₃$. Surface morphology analyzed using SEM showed that 0.2% Fe(III)-C/N-Bi₂O₃ have average diameter of 130 nm in the form of nanospheres. The absorption edge of $Bi₂O₃$ is around 475 nm with an energy gap of 2.61 eV. The Fe (III) grafting process and C/N doping did not change the Bi₂O₃ gap energy value. However, as a result of C/N doping and Fe(III) grafting displayed two peaks absorption in the visible light region So, C/N doping and Fe(III) grafting in principle can increase light absorption without changing the energy gap. The Fe(III)-C/N-Bi₂O₃ 0.2% photocatalyst had a degradation efficiency of 87% and showed excellent photocatalytic performance for decomposing 2.4-dichlorophenol under visible light with an irradiation duration of 60 minutes compared to pure $Bi₂O₃$ (24%).

The synthesis of Ce^{3+} doped on $Bi₂O₃$ was carried out by Zhang, W. and colle gues²⁴. XRD confirmed the crystallinity properties of the resulting material. Diffraction peaks of commercial nanoparticles $Bi₂O₃$ (CNB), hollow needleshaped $Bi₂O₃$ (HNB), and HNB-Ce-5 (5%) $CeO₂$ doping) identified were almost the same. Based on the resulting diffraction pattern, CNB and HNB were indicated as monoclinic α-Bi₂O₃ phases (JCPDS: 41-1449)²⁵. The peaks diffraction of $CeO₂$ doped on HNB (HNB-Ce), has the same pattern as pure $Bi₂O₃$. The phase structure of $Bi₂O₃$ did not change by dopant CeO₂ due to the small amount of dopant applied on $Bi₂O₃$ or some CeO₂ can enter the $Bi₂O₃$ lattice for similar ionic radii²⁶. The morphology of CNB an average size of 150 nm. HNB-Ce-5 had hollow needle-shape morphology where the size and surface were larger and rougher than CNB and HNB due to the presence of $CeO₂$ doping on HNB. The absorption spectrum of CNB has a strong absorption in the UV region. In contrast to CNBs, HNBs exhibit stronger light absorption in the visible light region due to their unique structure. However, HNB-Ce showed a higher light absorption property in the UV and visible light regions (200-800 nm). This suggests that a redshift from the absorption edge to a broader region can be attributed to $CeO₂$ doping²⁷. The energy gap of HNB-Ce is lower than that of CNB and HNB, it indicates that $CeO₂$ doping can reduce the energy gap of $Bi₂O₃$ and thereby increase absorption and utilization for visible light. The photocatalytic activity of the sample was evaluated by photodegradation of Tetracycline in a photochemical reactor and a lamp as a visible light source with an irradiation duration of 180 minutes. All HNB-Ce showed better photocatalytic properties than CNB and HNB. HNB-Ce-5 showed optimal photocatalytic properties. The degradation efficiency HNB-Ce-5 reached 89.1%, higher than CNB, HNB, HNB-Ce-3, and HNB-Ce-7 only reached 37.0%; 71.4%; 78.8%; and (76.9%), respectively.

Nagarajan, R. and colleagues²⁸ have stabilized β -Bi₂O₃ by doping 10% (%mol) thorium with solution combustion synthesis and co-precipitation methods. Diffraction peaks of the samples synthesized by the combustion method was indicated as monoclinic α -Bi₂O₃ phase (ICDD: 76-1730) for undoped samples. The monoclinic symmetry changes to tetragonal when the XRD pattern reveals that 10% Th⁴⁺ is doped on Bi^{3+} (ICDD: 78-1793). SEM image of sample β -Bi₂O₃ synthesized by the solution combustion method has a peeling-like morphology, while β -Bi₂O₃ synthesized by co-precipitation method shows a porous morphology with an average porous particle diameter of 1.7 nm. β -Bi₂O₃ which is synthesized by the co-precipitation method showed better efficiency due to a decrease in the energy gap. With 10% Th⁴⁺ in the $Bi₂O₃$ lattice using the combustion synthesis method, the energy gap decreases from 2.67 eV to 2.24 eV. However, synthesized by the co-precipitation method, the sample of 10% Th⁴⁺ doped on β -Bi₂O₃ has a drastic decrease of energy gap to 2.02 eV. Photodegradation activity was carried out on Methylene Blue and Rhodamine B under visible light irradiation. Samples are synthesized by the combustion method degrading up to 96% dye with a duration of irradiation for 120 minutes, while the samples are synthesized by the coprecipitation method only takes 90 minutes to degrading with the same percentage

degraded. The existence of carbon doping on Bi/Th^{4+} , which is synthesized by the coprecipitation method allows defects in the samples. This is a factor for increasing degradation efficiency in the samples.

Zhang, H. and colleagues²⁹, synthesized Bi₂O₃/CuNiFe Layered Double Hydroxide (LDHs) composites in the present study. Pure $Bi₂O₃$ XRD pattern matched with JCPDS: 1-071-2274 and the XRD pattern of CuNiFe LDHs based on JCPDS: 40-0215. Bi2O3/CuNiFe LDHs composites indicate all characteristic peaks as $Bi₂O₃$ peaks and do not shift. There are two peaks of diffraction of CuNiFe LDHs observed on the XRD spectrum for Bi₂O₃/CuNiFe LDHs composite. Morphology of CuNiFe LDHs like flower and layered, spherical $Bi₂O₃$, agglomerated nanoparticles due to the high surface energy of the $Bi₂O₃$ particles. The successfully synthesis of the $Bi₂O₃/CuNiFe$ LDHs composite was confirmed with there are many small nanoparticles scattered on the surface of the multi-layered CuNiFe LDHs. Such structures have the potential to increase photocatalytic activity of Bi2O3/CuNiFe LDHs composite. The absorption edge of $Bi₂O₃$ which was evaluated using UV-Vis spectrum is about 465 nm which corresponds to the intrinsic energy gap of $Bi₂O₃$ (2.82 eV). After combining, the composite LDH Bi2O3/CuNiFe showed stronger visible light absorption than LDH $Bi₂O₃$ pure CuNiFe, which indicates that the composite LDH $Bi₂O₃/CuNiFe$ has higher activity for pollutant degradation. It was proven that in sunlight, the degradation efficiency of Lomefloxacin by Bi2O3/CuNiFe LDHs was around 84.6% in 40 minutes of irradiation. Pure Bi2O³ and CuNiFe LDHs could only degrade 43.2% and 30.4%, respectively.

Wei, Z. et. al.³⁰, has succeeded in inserting BiOI nanosheets on the porous surface of $Bi₂O₃$, and it is evenly distributed. The XRD pattern is indicated as a tetragonal BiOI phase (JCPDS: 10-0445) and a monoclinic α -Bi₂O₃ (JCPDS: 41-1449). BiOI doping in $Bi₂O₃$ produced diffraction peaks similar to pure BiOI peaks. It cannot be ascertained clearly the peaks diffraction of $Bi₂O₃$, this is due to $Bi₂O₃$ being tightly encapsulated by BiOI. In addition, the prominent diffraction peaks of $Bi₂O₃$ and pure BiOI are very close, so overlapping each other can widen and weaken the XRD BiOI peaks. The characteristic peaks of Bi2O3/BiOI are wider than those of pure $Bi₂O₃$ and BiOI. BiOI's morphology has a diameter of about 1.0-1.5 μm and shows a form like a flower microsphere, and has a thickness of about 10-15 nm shaped nanosheet and see on the edge of the BiOI microsphere. The $Bi₂O₃$ sample showed 1D porous nanorod morphology with a 300-350 nm diameter. Meanwhile, the 50% Bi2O3/BiOI composite maintains a 1D nano rod-like structure with a 500-600 nm diameter. The $Bi_2O_3/BiOI$ composite shows strong light absorption in the visible light region than pure $Bi₂O₃$. The heterojunction structure by combining BiOI and $Bi₂O₃$ can increase the absorption of the visible light and reduce recombination rate of electron-hole pair that will increase degradation efficiency, other than the heterojunction structure is also able to expand the specific surface that contributes to the visible light absorption process. Significantly increased for Cr(VI) reduction under visible light irradiation compared to pure $Bi₂O₃$ and BiOI. In particular, a degradation rate of 94.5% can be achieved from 50% Bi2O3/BiOI composite with an irradiation duration of 100 minutes, while only 11.8% and 64.4% degradation efficiency by pure $Bi₂O₃$ and BiOI.

 $Cs_3PMo_{12}O_{40}/Bi_2O_3$ (CsPMo/Bi₂O₃) composite was synthesized by Wang, Qi and colleagues 31 . The results showed that $CsPMo$ successfully modified $Bi₂O₃$. The XRD pattern of the CsPMo/Bi₂O₃ composite contains three additional peaks with low intensity relative to pure $Bi₂O₃$ and are indicated as diffraction peaks of CsPMo. Doping CsPMo with a relativelly low concentration of 2.5% on the Bi_2O_3 surface causes the intensity of the CsPMo diffraction peaks that appears in the

 $CsPMo/Bi₂O₃$ composite. The pure $Bi₂O₃$ SEM image shows a flat ellipsoid or beams with a smooth surface, while the $CsPMo/Bi₂O₃$ composite has a rough surface. A relatively rougher surface is usually beneficial for photocatalysis because of the more active interfacial adsorption sites 32 . A redshift of the absorption edge after adding CsPMo to $Bi₂O₃$ indicated an increase in visible light absorption, which was more relative to pure $Bi₂O₃$. Compared to Bi_2O_3 (2.76 eV), the energy gap of $CsPMo/Bi₂O₃$ is smaller (2.63 eV), which is beneficial for the utilization of visible light. The photocatalytic activity of the samples was evaluated by Phenol degradation. $CsPMo/Bi₂O₃$ showed the highest activity after 300 minutes irradiation under visible light (83.6%) , while for Bi₂O₃ the degradation efficiency was around 48.0% and CsPMo only 12.5%.

Yakot, S. M.³³, also carried out the synthesis to form $β-Ni(OH)_2$ doped $α-Bi₂O₃$ composite materials. Each samples were synthesized by co-precipitation method, then the formation of $β-Ni(OH)_2$ doped α- $Bi₂O₃$ composite is done by mechanically mixed with different concentration of each samples. The XRD spectra showed that the monoclinic α -Bi₂O₃ phase and the β-Ni(OH)² hexagonal phase are well formed. Morphology of each samples are rods and sheets with diameters between 0.9 and 1.1 μm. The energy gap of pure α-Bi₂O₃ is 2.87 eV. While β -Ni(OH)₂ doped α -Bi₂O₃ composite (6-18 %wt) have an energy gap around 2.86-2.84 eV. β-Ni(OH)₂ doped $α Bi₂O₃$ which forms a composite resulted the red shift of absorption edge due to the interaction of $α$ -Bi₂O₃ after being mixed with β -Ni(OH)₂ is ascribed to the interface interaction between $α$ -Bi₂O₃ and β-Ni(OH)₂ particles. Modifying $α$ -Bi₂O₃ by β-Ni(OH)₂ was able to make more effective degrading. β -Ni(OH)₂ doped α -Bi₂O₃ composite showed degradation efficiency of 99% for Methylene Blue (80 min), 96% for Congo Red (80 min), 91% for Methyl Orange (180 min), and 90% for 4-nitrophenol (300 min) under visible light irradiation.

 $CaFe₂O₄-Bi₂O₃ heterojunction was$ successfully synthesized using the ultrasonic-assisted chemical co-precipitation method by Syed, A. and colleagues 34 . The position of the diffraction peaks was found to match those of $CaFe₂O₄$ and $Bi₂O₃$. The XRD pattern of pure $Bi₂O₃$ matches the structure of $Bi₂O₃$. The peak of the hematite phase is formed in the XRD pattern of $CaFe₂O₄$, it is due to the placement of Ca cations by replacing Fe cations in the crystal structure 35 . The XRD pattern of CaFe₂O₄- $Bi₂O₃$ showed similarities to both the peaks of $CaFe₂O₄$ and $Bi₂O₃$ nanoparticles. Morphological analysis using TEM described CaFe2O⁴ nanoparticles in nanospheres on the surface of $Bi₂O₃$ nanosheets and spread evenly without any aggregation. The photon absorption ability of $CaFe₂O₄ - Bi₂O₃$ nanocomposite is in the energy gap region of 2.16 eV. The photodegradation efficiency of $CaFe₂O₄$ -Bi2O³ was evaluated on Methylene Blue dye. A decrease in the concentration of Methylene Blue was observed under visible light irradiation for 180 minutes. The $CaFe₂O₄ - Bi₂O₃$ nanocomposite showed an 8 to 16-fold increase in the kinetic rate constant than Bi_2O_3 and $CaFe_2O_4$ for the degradation of Methylene Blue dye. The $CaFe₂O₄-Bi₂O₃$ nanocomposite is magnetically recoverable with high reusable capacity.

The $Bi_2O_3/FeVO_4$ heterojunction semiconductor was prepared by Liu, X. and Y. Kang³⁶. XRD pattern of $Bi₂O₃$ has a monoclinic α -Bi₂O₃ phase (JCPDS: 41-1449) and all diffraction peaks of $FeVO₄$ are confirmed by JCPDS: 38-1372. Meanwhile, diffraction peaks of Bi2O3/FeVO⁴ heterojunction shows the diffraction peaks of both the crystalline phase of $Bi₂O₃$ and FeVO₄ which proves that the $Bi₂O₃/FeVO₄$ heterojunction has been successfully synthesized. A solid and sharp $Bi₂O₃$ peak indicates a high degree of crystallinity, while a low FeVO₄ peak indicates a low degree of crystallinity. The absorption spectrum of the $Bi₂O₃/FeVO₄ hetero,$ is around 500 nm to 700 nm, this indicates

that samples are able to beneficial utilization of visible light more effective and efficient. The energy gap of $Bi_2O_3/FeVO_4$ heterojunction is 2.08 eV, $Bi₂O₃$ is 2.86 eV, and $FeVO₄$ is 2.26 eV. It shows that the $Bi₂O₃/FeVO₄$ heterojunction can easily induce more electrons and holes by visible light. The photocatalytic activity was tested by degrading Malachite Green for 240 minutes under visible light irradiation. Efficiency degradation of the $Bi_2O_3/FeVO_4$ heterojunction was 88.7% higher than pure $Bi₂O₃$ (67.9%) and FeVO₄ (58.7%).

Ramachandran, S. and A. Sivasamy³⁷, carried out a synthesis to produce $ZnO/Bi₂O₃$ composites by precipitation and ultrasonication methods. The crystallinity of the resulting $ZnO/Bi₂O₃$ nanomaterial with a ratio of 3:1 (ZB3) showed the hexagonal wurtzite structure of ZnO (JCPDS: $36-1451$)³⁸. In addition, the peak pattern in Bi₂O₃ corresponds to the α-Bi2O³ phase monoclinic (JCPDS: 6-294) and tetragonal β-Bi2O³ phase (JCPDS: 27- $50³⁹$. The results of FE-SEM analysis confirmed that the Bi_2O_3 surface are rodshaped with micro-size. The ZnO nanoparticles structures look typical and no agglomeration occurs. Increased contact between the surface of Bi_2O_3 and ZnO due no agglomeration will increases photocatalytic activity. The optical properties of the nanomaterials were analyzed using UV-Vis DRS (Diffuse Reflectance Spectroscopy) and the results show ZB3 can absorbs light in the visible spectrum with an energy gap of 3.12 eV. The composite energy gap is lower than ZnO. The shift of absorption peak due to ZnO modification with $Bi₂O₃$ can extended its photoactivity. The nano photocatalyst property of ZB3 was explored by conducting experiments on the photocatalytic degradation of Acid Red-85 dye under visible light irradiation. After 240 minutes irradiation, ZB3 degradation efficiency reached 93.53%

Bi2O3-bentonite nanocomposite was successfully synthesized by Patil, S. P. and colleagues^{40} to degrade Rhodamine B. The results of XRD analysis showed that all the Bi2O3-bentonite diffraction patterns were similar to the $Bi₂O₃$ diffraction patterns⁴¹, but there are peaks widening on $Bi₂O₃$ bentonite compared to pure $Bi₂O₃$. The morphology of the samples analyzed using SEM showed that $Bi₂O₃$ has the form of nanorods, it is clear that the $Bi₂O₃$ nanorods are well dispersed on the bentonite. Bentonite, $Bi₂O₃$, and $Bi₂O₃$ -bentonite are used for degrading of Rhodamine B with a duration of irradiation in visible light for 80 minutes. The percentage of Rhodamine B degradation by bentonite, $Bi₂O₃$, and $Bi₂O₃$ bentonite after 80 minutes of adsorption were 62%, 58.4% and 98.5%, respectively. Intercalation between bentonite and $Bi₂O₃$, increasing light absorption and decreasing electron-hole pair recombination can increase photocatalytic efficiency than pure $Bi₂O₃$.

Zhu, W, and colleagues⁴¹, synthesized Bi2O3/CuO composites using the co-precipitation method. Based on XRD analysis, the CuO diffraction peak was not detected, but what was detected was the Bi7.38Cu0.62O11.69 peak (JCPDS: 00–049- 1765). A new compound replaced the disappearance of the CuO peak. On the other hand, the main feature of $Bi₂O₃$ is still maintained, which is confirmed by JCPDS: 03–065-2366. Bi_{7.38}Cu_{0.62}O_{11.69} grows on the surface of $Bi₂O₃$. The results of SEM analysis of pure $Bi₂O₃$ have irregular shapes and sizes, the particle size ranging from 40 nm to 200 nm. CuO doped on $Bi₂O₃$ has a morphology like seedlings (CuO) that grows on the surface of $Bi₂O₃$ and forms nanoparticles. Pure $Bi₂O₃$ and $Bi₂O₃/CuO$ composites have a strong absorption capacity for UV light. At a wavelength of 380 nm, the absorption curve of pure $Bi₂O₃$ decreases sharply, but Bi_2O_3/CuO composites show superiority with higher absorption in the visible light region. The absorption rate of $Bi₂O₃/CuO$ is higher, which is more than three times that of pure $Bi₂O₃$. The results showed that the addition of CuO extended the absorption range of the photocatalyst from ultraviolet to visible

light. The gap energies of $Bi₂O₃$ and Bi2O3/CuO composites are 2.8 eV and 1.9 eV, respectively. The presence of dopants determines the decrease in the energy gap; photon energy utilization is positively affected. Compared with pure $Bi₂O₃$, Bi2O3/CuO nanocomposite showed higher catalytic reaction and photocatalytic efficiency against the target pollutant tetracycline hydrochloride by irradiation by visible light, the obtained degradation efficiency of Bi2O3/CuO composite was 97.22%, while the efficiency of pure $Bi₂O₃$ was 35.12%. After four repeated experimental cycles, the degradation efficiency of $Bi₂O₃/CuO$ composites can still reach more than 90%. This proves that the material has repeated stability for Tetracycline Hydrochloride degradation. These characteristics make $Bi₂O₃/CuO$ composites have practical application value.

Xie, T. and colleagues^{42} , conducted a study to synthesize SrFe₁₂O₁₉ doped β-Bi2O³ as a magnetic photocatalyst. Diffraction peaks of $SrFe_{12}O_{19}$ (15%) doped $β$ -Bi₂O₃ were confirmed as $β$ -Bi₂O₃ phase (JCPDS: 27-0050). Diffraction peak from β- $Bi₂O₃$ on SrFe₁₂O₁₉ doped β-Bi₂O₃ was still strong, which indicated that additional of doping did not change phase of β-Bi2O3. The result of the XRD pattern analysis confirmed that diffraction peaks of $SrFe₁₂O₁₉$ did not observed due concentration of is low. Morphological analysis of SrFe₁₂O₁₉ doped β-Bi₂O₃ was performed using SEM. SrFe $_{12}O_{19}$ has a micron particles structure. Based on JCPDS: $24-1207$, SrFe₁₂O₁₉ has a hexagonal crystal system and it is proven from the results of SEM analysis that the hexagonal crystals are perfectly formed and the crystal planes grow uniformly. The addition of $SrFe_{12}O_{19}$ had no significant effect, it indicates that composite are well dispersed and heterojunctions structure can be formed which has the potential to make $β$ -Bi₂O₃ more effective in degrading. The results of UV-VIS DRS indicate that absorbance spectrum of SrFe₁₂O₁₉ doped β -Bi₂O₃ composite has a strong absorption than β -Bi₂O₃ in visible light region. The addition of SrFe₁₂O₁₉ to β- $Bi₂O₃$ can decreased of gap energy to 2.38 eV, confirming that SrFe12O¹⁹ doping could decrease the energy gap and thereby extend the absorbance range. The photocatalytic activity was tested against Rhodamine B using β -Bi₂O₃ and SrFe₁₂O₁₉ doped β -Bi₂O₃ composite. The photocatalytic activity increased after doping SrFe12O¹⁹ 0-15% and the most optimal was obtained at $S_fFe_{12}O_{19}$ 15% doping on β -Bi₂O₃ (92.97%) with a visible light irradiation duration of 150 minutes. However, additional 20-25% can

decreased photocatalytic efficiency even worse than that of pure β -Bi₂O₃. The degradation rate of pure β-Bi₂O₃ reached 71.32%, while 68.74% for 25% SrFe₁₂O₁₉. The gap energy measurement showed that SrFe₁₂O₁₉ doped β -Bi₂O₃ composite (25%) has the lowest value. In the theory, the lower gap energy can absorb more visible light but this does not applied, because doping up to 25% causes a decrease of β- $Bi₂O₃$ content. [Table 1] presents a summary of the photocatalytic activity of the doped Bi₂O₃.

PHOTOCATALYSIS MECHANISM

Photocatalysts can be defined as materials that can speed up reactions with the help of light energy. The process that occurs in photocatalyst is called photocatalysis. Almost all photocatalyst materials are semiconductor materials because they have an energy gap of about 1- 4 eV. The characteristics of an ideal photocatalyst material are that photons can activate it, is chemically unreactive, nontoxic, easy to obtain, and able to utilize a broad spectrum of sunlight⁴³. [Fig 2] shows

the mechanism of the photocatalytic process.

Pairs of electrons (e^-) and holes (h^+) play a role in determining the reaction process that will take place in both oxidation and reduction states. The photocatalytic process occurs via direct charge transfer consisting of photoinduction carriers or reactive oxygen species $(ROS)^{44}$. Electrons in the valence band will be excited when exposed to sunlight, the process is called the reduction process⁴⁵. Electrons that do not recombine and succeed to the semiconductor surface will adsorb $O₂$

molecules to form compounds ${}^{\bullet}O_2^$ or superoxide anion radicals, reducing species. When a substance molecule meets a hole (h⁺), the process is the oxidation process. The hole formed in the valence band will act as an oxidizing agent for H_2O . This hole will react with OH⁻ which is adsorbed on the surface of the semiconductor to form a hydroxyl radical compound ('OH) which is a powerful oxidizing agent and the hole from visible light acts as an oxidizer to become O_2 . These reducing and oxidizing species can attack contaminants in water and degrade into harmless compounds⁴⁶.

Constraints that often occur in semiconductor photocatalysts are the high rate of electron-hole pair recombination so that the electron lifetime becomes shorter, this affects photocatalytic activity. In the presence of doping, it will be responsible for the electron-hole pair recombination rate and support the increase in photocatalytic activity by suppressing the electron-hole pair recombination rate. However, excessive ion or doping material makes it enter the cluster formation. This cluster can restrain the rate of photodegradation by covering the active site from the surface of $Bi₂O₃$. Doping material acts as photo-generated between holes and electron transfer, the rate of electron-hole pair recombination during irradiation can be suppressed by increasing the number of trapped electrons to increase the lifetime of electrons and holes. This decrease in the recombination rate increases the photocatalytic activity of $Bi₂O₃$.

DISCUSSION AND PERSPECTIVE

The dopant $Bi₂O₃$ photocatalyst increased its photocatalytic activity due to surface modification, widening of the absorption region towards the visible region, and a decrease in the energy gap of the material. The precipitation method carried out the synthesis to produce a powdered material. The use of photocatalysts from granular materials produces residues in the aquatic environment. Therefore, according to the authors, in addition to developing $Bi₂O₃$ materials that are effective and efficient in degrading, it is also important to consider application techniques that are more environmentally friendly (not causing new problems, such as heavy metal residues).

CONCLUSION

Generally, the photocatalytic activity of semiconductor photocatalysts depends on several factors such as the crystal structure, morphology, surface area, and electronic structure. Bi₂O₃ photocatalyst doped with elements or compounds to form a composite with precipitation synthesis method can modify pure $Bi₂O₃$, so that its photocatalytic activity increases. The addition of dopants to $Bi₂O₃$ with the appropriate concentration will not change the crystalline phase of $Bi₂O₃$, the surface morphology formed is relatively rougher, and the absorption area is more comprehensive to the visible light region.

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